Home Search Collections Journals About Contact us My IOPscience

A systematic study of four series of electron-doped rare earth manganates, $Ln_x Ca_{1-x} MnO_3$ (Ln = La, Nd, Gd and Y) over the x = 0.02–0.25 composition range

This article has been downloaded from IOPscience. Please scroll down to see the full text article. 2003 J. Phys.: Condens. Matter 15 895 (http://iopscience.iop.org/0953-8984/15/6/316)

View the table of contents for this issue, or go to the journal homepage for more

Download details: IP Address: 171.66.16.119 The article was downloaded on 19/05/2010 at 06:34

Please note that terms and conditions apply.

J. Phys.: Condens. Matter 15 (2003) 895-905

A systematic study of four series of electron-doped rare earth manganates, $Ln_xCa_{1-x}MnO_3$ (Ln = La, Nd, Gd and Y) over the x = 0.02-0.25 composition range

L Sudheendra, A R Raju and C N R Rao¹

Chemistry and Physics of Materials Unit, Jawaharlal Nehru Centre for Advanced Scientific Research, Jakkur PO, Bangalore-560064, India

E-mail: cnrrao@jncasr.ac.in

Received 4 September 2002, in final form 11 November 2002 Published 3 February 2003 Online at stacks.iop.org/JPhysCM/15/895

Abstract

Electrical and magnetic properties of four series of manganates Ln_xCa_{1-x}MnO₃ (Ln = La, Nd, Gd and Y) have been studied in the electron-doped regime (x = 0.02-0.25) in order to investigate the various inter-dependent phenomena such as ferromagnetism, phase separation and charge ordering. The general behaviour of all four series of manganates is similar, with some of the properties showing a dependence on the average radius of the A-site cations, $\langle r_A \rangle$ and cation size disorder. Thus, all the compositions show an increase in magnetization at 100–120 K (T_M) for $x < x_{max}$, the magnetization increasing with increasing x. The value of x_{max} increases with decreasing $\langle r_A \rangle$, probably due to the increased phase separation induced by site disorder. This is also reflected in the larger width of the hysteresis loops at $T < T_M$ for small x or $\langle r_A \rangle$. In this regime, the electrical resistivity decreases with increasing x, but remains low and nearly constant, $T > T_M$. The percolative nature of the conduction mechanism at $T < T_M$ is substantiated by the fit of the conductivity data to the scaling law, $\sigma \propto |x_c - x|^p$, where p is in the 2–4 range. When $x > x_{max}$, the materials become antiferromagnetic (AFM) and charge-ordered at a temperature T_{CA} , accompanied by a marked increase in resistivity. The value of T_{CA} increases with increasing $\langle r_A \rangle$ and x (up to x = 0.3). Thus, all four series of manganates are characterized by a phase-separated regime between x = 0.02 and 0.1–0.15 and an AFM charge-ordered regime at x > 0.1–0.15.

1. Introduction

The electron-doped regime of rare earth manganates of the general formula $Ln_x Ca_{1-x} MnO_3$ (Ln = rare earth, x < 0.5) have attracted considerable attention because of the unusual properties exhibited by the compositions in this regime [1, 2]. The electron-doped manganates

0953-8984/03/060895+11\$30.00 © 2003 IOP Publishing Ltd Printed in the UK

¹ Author to whom any correspondence should be addressed.

are different from the hole-doped manganates (x > 0.5) in several ways, and electron-hole asymmetry in the manganate system has been a subject of discussion [2]. The electron-doped manganates are dominated by charge ordering and do not exhibit the ferromagnetic (FM) ground state at any composition. Even small doping in the A-site ($x \le 0.03$) has marked effects on the electronic properties of the manganates [3]. In the $Ln_xCa_{1-x}MnO_3$ system, compositions with x < 0.03 exhibit G-type antiferromagnetism while those with x = 0.1-0.15shows C-type antiferromagnetism, often in admixture with G-type antiferromagnetism. FM clusters are present in the entire antiferromagnetic (AFM) regime [4, 5]. Compositions in the x = 0.03-0.1 range show a significant increase in magnetization around 100 K due to the presence of FM clusters in the AFM matrix. There is also a change in the resistivity behaviour around 100 K and the $x \leq 0.1$ compositions exhibit low electrical resistivity above this temperature [6–8]. Magnetic susceptibility and resistivity data show some history dependence or irreversibility as well [5]. Charge ordering appears to manifest itself when x > 0.1-0.15, but the manner of evolution of this phenomenon with changes in temperature and composition has not been fully examined. It is generally believed that there is phase separation at low temperatures, with the size of the FM domains increasing with the dopant concentration in the x = 0.02-0.1 composition range [5, 9]. Electrical conduction in this regime could be percolative.

In spite of several studies on the electron-doped regime of the $Ln_x Ca_{1-x} MnO_3$ system reported in the literature, there are many aspects of these materials that we do not understand fully. Some important aspects of interest are the following: is the FM cluster or spin-glass regime (x < 0.1-0.15), where the magnetization increases with x or electron concentration, sensitive to the average A-site cation radius, $\langle r_A \rangle$? Is the resistivity behaviour metallic at temperatures beyond the spin-glass regime? In what way is the phase separation dependent on the composition and $\langle r_A \rangle$? Does size disorder play a role? Is the conduction percolative at low temperatures for x < 0.1-0.15 and, if so, is it affected by the A-site cation radius or site disorder? When does the charge ordering manifest itself and what is the effect of the A-site cation radius on the charge-ordering transition? In order to contribute to the understanding of some of these issues, we have carried out systematic resistivity and magnetization measurements on $Ln_xCa_{1-x}MnO_3$ (Ln = La, Nd, Gd and Y) with x varying between 0.02 and 0.25, making sure that there are sufficient compositions in the x < 0.1 and x > 0.1 regimes. The latter regime (x = 0.1-0.25) is relevant to examine the emergence of charge-ordering effects. Although measurements on some of these manganate compositions are reported in the literature, by Neumeier et al [3, 4], Raveau and co-workers [5–9] and others, it was our considered view that careful measurements were necessary on a related series of manganates, covering a range of compositions from the pure G-AFM regime at very low x to the charge-ordered regime at x = 0.1–0.25, and with different Ln substitutions, in order to fully understand the electron-doped regime. The present study throws some light on the various phenomena such as phase separation, charge ordering and percolative conduction occurring in the x = 0.02-0.25 range of the electron-doped rare earth manganates and helps to adequately describe the characteristics of this fascinating regime.

2. Experimental details

Polycrystalline powders of $Ln_x Ca_{1-x} MnO_3$ (Ln = La, Nd, Gd and Y) compositions were prepared by the solid-state reaction of stoichiometric amounts of the rare earth acetate with CaCO₃ and MnO₂. The starting materials were ground and heated to 1000 °C for 60 h with three intermediate grindings. Then the samples were reheated at 1200 °C for 48 h with two intermediate grindings. The samples were then pelletized and heated to 1325 °C for 36 h.



Figure 1. Variation of (a) $\langle r_A \rangle$ and (b) σ^2 with *x* with $Ln_x Ca_{1-x} MnO_3$.

X-ray powder diffraction measurements were carried out using a Seifert 3000 diffractometer. All the manganate compositions had an orthorhombic structure, the lattice parameters varying with the Ln ion in the $\text{Ln}_x \text{Ca}_{1-x} \text{MnO}_3$ (x = 0.02-0.25) compositions. Thus, the lattice parameters decrease as we go from La to Y, but within a Ln series there is only a small variation with increase in x. $\langle r_A \rangle$ decreases with increase in x (except when Ln = La) in each series (figure 1(a)). It is to be noted that the site disorder, as measured by the variance σ^2 [10], also varies with x and the Ln ion. When Ln = La or Nd the σ^2 is relatively small, varying little with x, but when Ln = Gd or Y, σ^2 becomes appreciable, increasing with x as shown in figure 1(b). The σ^2 is maximum and $\langle r_A \rangle$ is smallest in the yttrium system (figure 1). The samples were analysed for their elemental compositions by employing EDX analysis with quantitative ZAF correction software. Electrical resistivity measurements were carried out at 4000 G using a LakeShore 7300 vibrating sample magnetometer.



Figure 2. Temperature variation of magnetization of $Ln_x Ca_{1-x} MnO_3$ with x = 0.02-0.25: (a) Ln = La, (b) Ln = Nd.

3. Results and discussion

In figures 2 and 3 we show the temperature variation of magnetization for several compositions of $Ln_x Ca_{1-x}MnO_3$ with Ln = La, Nd, Gd and Y. The compositions with $x \leq 0.1$ in all four series of manganates exhibit a marked increase in magnetization as in a ferromagnet around 100–120 K (T_M). These materials are, however, not real ferromagnets and show only small values of saturation magnetization even at 9000 G. At low x values ($x \leq 0.1$), the magnetization increases with x or the electron concentration and then decreases sharply. The maximum value of magnetization occurs at x = 0.08, 0.10 and 0.15 (x_{max}) for Ln = La, Nd, Gd and Y, respectively. Accordingly, plots of μ_β versus x show maxima at increasing values of x as the average radius of the A-site cation, $\langle r_A \rangle$, decreases, the maximum value of μ_β



Figure 3. Temperature variation of magnetization of $Ln_x Ca_{1-x} MnO_3$ with x = 0.02-0.25: (a) Ln = Gd, (b) Ln = Y.

being found when Ln = Gd (figure 4). The increase in x_{max} with decrease in $\langle r_A \rangle$ can arise from phase separation due to the presence of significant FM fractions in the AFM matrix. The largest FM fraction occurs around x_{max} when the magnetization is maximum. In the composition range x > 0.15 ($x > x_{max}$), the FM fraction is small, with the concentration decreasing with increasing x. We would therefore expect phase separation to be prominent up to x_{max} . Considering that ferromagnetism itself would be favoured by large $\langle r_A \rangle$, the observed trend in figure 4 can be taken to reflect an increase in the width of the phase separation regime. Interestingly, the x_{max} is highest when Ln = Y. This is likely to be because of the large cation size disorder (figure 1). This observation indirectly suggests that the phase separation in these manganate compositions is induced by size disorder, the separation regime increasing with σ^2 .

After the magnetization attains a maximum value at x_{max} , we not only see a sudden drop in magnetization, but also evidence for competition between ferromagnetism and

Figure 4. Variation of μ_{β} with composition at 50 K for Ln_xCa_{1-x}MnO₃.

Figure 5. Variation of the AFM charge-ordering transition temperature. T_{CA} with the average A-site cation radius $(\langle r_A \rangle)$ in Ln_xCa_{1-x}MnO₃ for different values of x ($x > x_{max}$).

antiferromagnetism. This competition gives rise to a peak in the magnetization–temperature curves (see figures 2 and 3). These peaks are reminiscent of the magnetization peaks found in charge-ordered systems such as Nd_{0.5}Ca_{0.5}MnO₃ and Pr_{0.6}Ca_{0.4}MnO₃ [1, 2]. The temperature corresponding to the peak maximum (T_{CA}) increases with increasing x in the x = 0.1-0.25 composition range in all four series of manganates. The peak occurs at $x \sim 0.1$ when Ln = La, at 0.15 for Nd and Gd and at 0.2 for Y. As shown in figure 5, T_{CA} increases with $\langle r_A \rangle$. It has

Figure 6. Temperature variation of the electrical resistivity of $Ln_x Ca_{1-x} MnO_3$ with x = 0.02-0.25: (a) Ln = La, (b) Ln = Nd.

been shown that the temperature corresponding to the magnetic susceptibility anomaly due to charge ordering in electron-doped manganate compositions increases up to x = 0.3 and then decreases slightly in the x = 0.3–0.5 composition range [2, 11]. The effect of $\langle r_A \rangle$ is negligible when $x \ge 0.3$.

The changes in the magnetization of $Ln_x Ca_{1-x} MnO_3$ with composition and temperature are reflected in the electrical resistivities [6] as shown in figures 6 and 7. Thus, all four series of manganates with $x < x_{max}$ show low resistivities from 300 K down to 100–120 K (T_M), independent of $\langle r_A \rangle$. The activation energies for conduction are rather small (25– 30 meV), as reported by other workers [7, 8]. In this low dopant concentration regime ($x < x_{max}$), the resistivity increases below T_M , the change possibly representing a semi-metal–

Figure 7. Temperature variation of the electrical resistivity of $Ln_x Ca_{1-x} MnO_3$ with x = 0.02-0.25: (a) Ln = Gd, (b) Ln = Y.

insulator transition. The low-temperature resistivity (at $T < T_M$) increases with the decrease in electron concentration or x in this composition regime, paralleling the magnetization data, in agreement with the earlier literature [6]. The resistivity also decreases with the decrease in $\langle r_A \rangle$ in the Ln_xCa_{1-x}MnO₃ series, showing a minimum for Ln = Gd ($\langle r_A \rangle \sim 1.179$ Å) where the magnetization is maximum. Such a correspondence between the magnetization and the resistivity data is interesting and may have its origin in the phase separation resulting in percolative conduction. We shall examine this aspect later in the paper.

The resistivity behaviour drastically changes when $x > x_{max}$ in the four series of manganates. In this composition regime, the resistivity increases sharply, as can be seen from figures 6 and 7. The increase in resistivity occurs around the same temperature as the peak in the magnetization–temperature curves (T_{CA}). The occurrence of a sharp change in resistivity at the same temperature as the magnetization peak suggests the occurrence of charge ordering associated with antiferromagnetism. This transition temperature may, therefore, be considered to represent the onset of C-type antiferromagnetism.

The $x < x_{max}$ compositions in $Ln_xCa_{1-x}MnO_3$ exhibit magnetic hysteresis at $T < T_M$ (figure 8). The hysteresis loops throw light on the nature of phase separation. In the multilayers

Figure 8. Magnetic hysteresis in $Ln_x Ca_{1-x} MnO_3$ (x = 0.02, 0.04, 0.06 and 0.08) at 50 K.

of spin valve and permalloy materials [12–14], wherein the FM layers are coupled to AFM layers, the hysteresis loops reflect the extent of exchange coupling between the FM and the AFM layers. The coercivity of the FM layer increases due to the coupling with an AFM layer. Since the phase-separated compositions of the manganates contain both the FM and the AFM regions, they would be expected to show similar behaviour. We see from figure 8 that the hysteresis loops are broad at small x values ($x \le 0.04$) and the width decreases as x reaches x_{max} . When the system is subjected to a magnetization reversal process, the interfacial spins between AFM and FM domains, leading to broadening of the hysteresis loops. As the electron concentration or x increases, the FM character (domain/cluster size) increases, thereby reducing the drag substantially and causing a decrease in the width, as expected. It must be recalled that the FM fraction reaches a maximum at x_{max} . The width of the hysteresis loop for a given x value increases with the decrease in the radius of the Ln (or $\langle r_A \rangle$), reflecting the effect of phase separation.

The FM hysteresis loops are symmetric, indicating two equivalent directions of magnetization. On the other hand, when a FM/AFM material is cooled in an external magnetic field (as in this case for obtaining M versus T plots) below the Néel temperature (<100 K), the loop is shifted from zero due to the exchange bias effect. The small differences in the coercive fields (table 1) seen for the phase-separated samples may be due to the pinning of a small fraction of FM interfacial spins to the AFM domains. The pinned spins do not rotate in an external field as they are coupled, leading to exchange biasing.

Magnetization reversal in an FM system occurs through rotation of spins, and via domain nucleation and growth. An examination of the hysteresis loops shows that magnetization drops from a maximum value to zero sharply compared to the increase from zero to the maximum

Figure 9. Conductivity data of $\text{Ln}_x \text{Ca}_{1-x} \text{MnO}_3$ at 50 K fitted to the scaling law $\sigma \propto |x_c - x|^p$. The inset shows the conductivity data fitted to the equation at 20 K.

Table 1. Coercivity (H_c and $-H_c$ in gauss) of $Ln_x Ca_{1-x} MnO_3$ obtained from hysteresis measurements at 50 K.

	La		Nd		Gd		Y	
x	H_c	$-H_c$	H_c	$-H_c$	H_c	$-H_c$	H_c	$-H_c$
0.02	966	849	454	340	508	395	429	338
0.04	706	604	454	449	580	572	865	589
0.06	300	203	215	124	373	279	579	473
0.08	94	100	114	32	154	62	344	247
0.10		_	870	776	154	62	322	220
0.15	—	—	—	—	—	—	310	208

value. This suggests that, for $0.02 < x \le 0.06$, one mechanism dominates over the other in a particular region of magnetization reversal process, bringing about an asymmetry in the hysteresis loops. Thus, it is much harder to magnetize the $0.02 < x \le 0.06$ compositions wherein the FM clusters are embedded in a AFM matrix.

It was mentioned earlier that the conducting mechanism in the phase-separated regime may be percolative. We have employed the scaling law, $\sigma \propto |x_c - x|^p$, to treat the resistivity data in the x = 0.02–0.1 composition regime. We show $\ln \sigma - \ln x$ plots at 50 and 20 K from the series of manganates studied by us in figure 9. The value of the exponent p is between 2.1 and 2.8 at 50 K and between 2.8 and 4.3 at 20 K. Percolative conduction becomes less dominant as the temperature approaches T_M or x reaches x_{max} (0.1–0.15). This is understandable since antiferromagnetism is the main interaction when $x > x_{max}$ and $T > T_M$, leading to lesser phase separation and a more homogeneous AFM phase.

4. Conclusions

The present study of the four series of electron-doped manganates, $Ln_x Ca_{1-x} MnO_3$ (Ln = La, Nd, Gd and Y), over a wide range of compositions (x = 0.02-0.25) has been useful in understanding the evolution of various phenomena. Thus, these materials, which show FMlike behaviour at $T < T_M$ up to a value of x (x_{max}), become AFM with charge ordering at T_{CA} for $x > x_{max}$. The values of x_{max} and T_{CA} depend on the average radius of the A-site cation $(\langle r_A \rangle)$, the former being related to the phase separation and site disorder (σ^2). Phase separation is favoured at low temperatures $(T < T_M)$ by small $x (\langle x_{max} \rangle)$ and small $\langle r_A \rangle$ (or large σ^2). The AFM CO transition temperature, T_{CA} , on the other hand, increases with increasing x and $\langle r_A \rangle$ and does not vary significantly with $\langle r_A \rangle$ when $x \ge 0.3$. All these materials show low resistivity at $T > T_M$ when $x \leq x_{max}$, but show a sharp increase in resistivity at T_{CA} when $x > x_{max}$, due to charge ordering. When $x < x_{max}$ and $T < T_M$, conduction appears to be percolative. Since the above features are found in all four series of manganates covering a wide range of $\langle r_A \rangle$, they can be taken to be intrinsic to the electron-doped compositions of the rare earth manganates. Thus, the present study clearly identifies the phase-separated regime with percolative conduction ($x = 0.02 - x_{max}$) and the charge-ordered AFM regime with x_{max} in the $\sim 0.1-0.15$ range. The percolative regime increases with size disorder or decreases in $\langle r_A \rangle$.

References

- [1] Rao C N R, Arulraj A, Cheetham A K and Raveau B 2000 J. Phys.: Condens. Matter 12 R83
- [2] Vijaya Sarathy K, Vanitha P V, Seshadri R, Cheetham A K and Rao C N R 2001 Chem. Mater. 13 787
- [3] Granado E, Moreno N O, Martinho H, Garcia A, Sanjurjo J A, Torriani I, Rettori C, Neumeier J J and Oseroff S B 2001 Phys. Rev. Lett. 86 5385
- [4] Neumeier J J and Cohn J L 2000 Phys. Rev. B 61 14319
- [5] Mahendiran R, Maignan A, Martin C, Hervieu M and Raveau B 2000 Phys. Rev. B 62 11644
- [6] Maignan A, Martin C, Damay F and Raveau B 1998 Chem. Mater. 10 950
- [7] Maignan A, Martin C, Damay F, Raveau B and Hejtmanek J 1998 Phys. Rev. B 58 2758
- [8] Hejtmanek J, Jirak Z, Marysko M, Martin C, Maignan A, Hervieu M and Raveau B 1999 *Phys. Rev.* B 60 14057
 [9] Martin C, Maignan A, Hervieu M, Raveau B, Jirak Z, Savosta M M, Kurbakov A, Trounov V, Andre G and Bouree F 2000 *Phys. Rev.* B 62 6442
 - Respaud M, Broto J M, Rakoto H, Vanacken J, Wagner P, Martin C, Maignan A and Raveau B 2001 *Phys. Rev.* B **63** 144426
- [10] Rodriguez-Martinez L M and Attfield J P 1996 Phys. Rev. B 54 R15622
- [11] Santosh P N, Aruraj A, Vanitha P V, Singh R S, Sooryanarayana K and Rao C N R 1999 J. Phys.: Condens. Matter 11 L27
- [12] Meiklejohn W H and Bean C P 1956 Phys. Rev. 102 1413
- [13] Dimitrov D V, Zhang S, Xiao J Q, Hadjipanayis G C and Prados C 1998 Phys. Rev. B 58 12090
- [14] Li Z and Zhang S 2000 Phys. Rev. B 61 R14897